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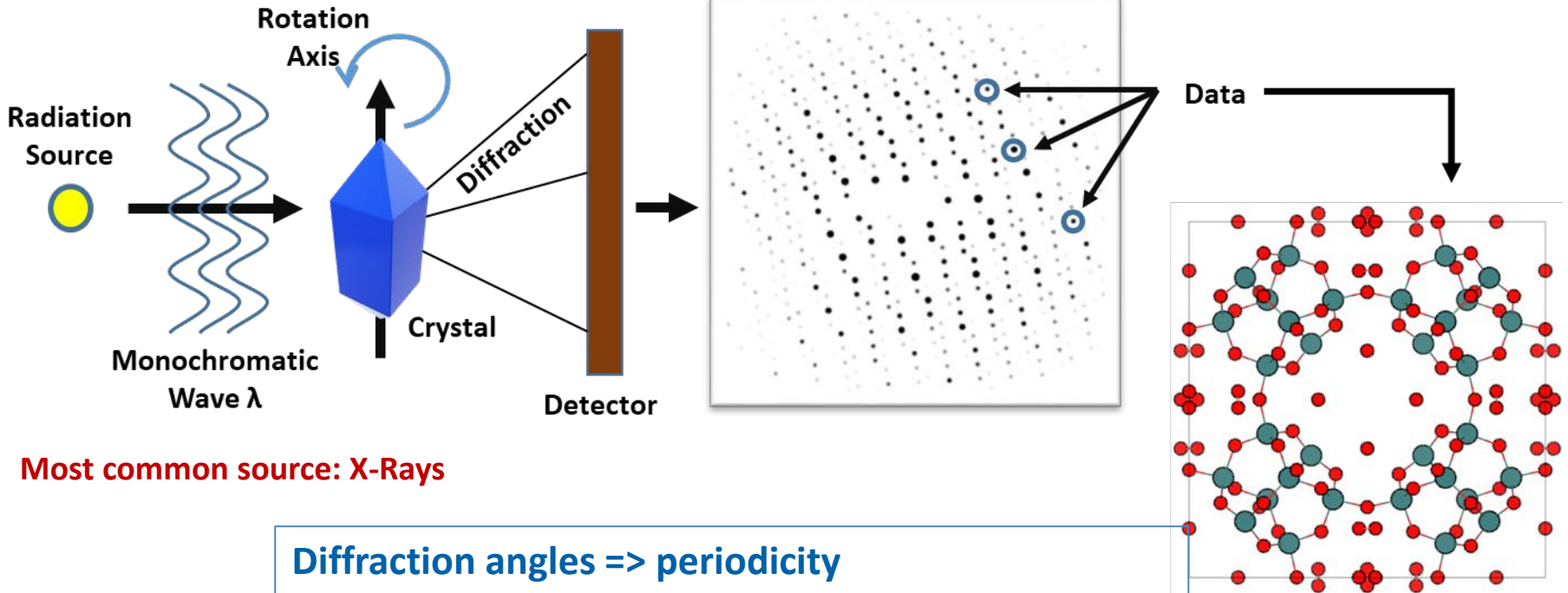
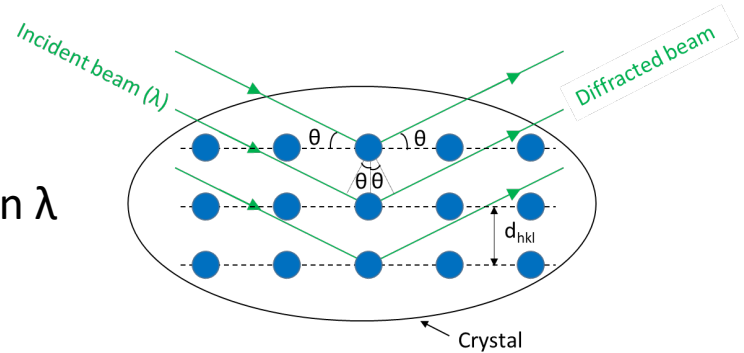
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# Transmission electron microscopy in materials science: Advances in electron crystallography

New Inorganic Functional Oxides: Synthesis, Characterisation and Simulations  
Orléans, 4-6 october, 2023

# Investigating the structure of matter: diffraction experiment

Bragg's law:  $2 d_{hkl} \sin \theta = n \lambda$



Most common source: X-Rays

Diffraction angles => periodicity  
 Intensities => motif.  $I(hkl)$  proportional to  $|F(hkl)|^2$

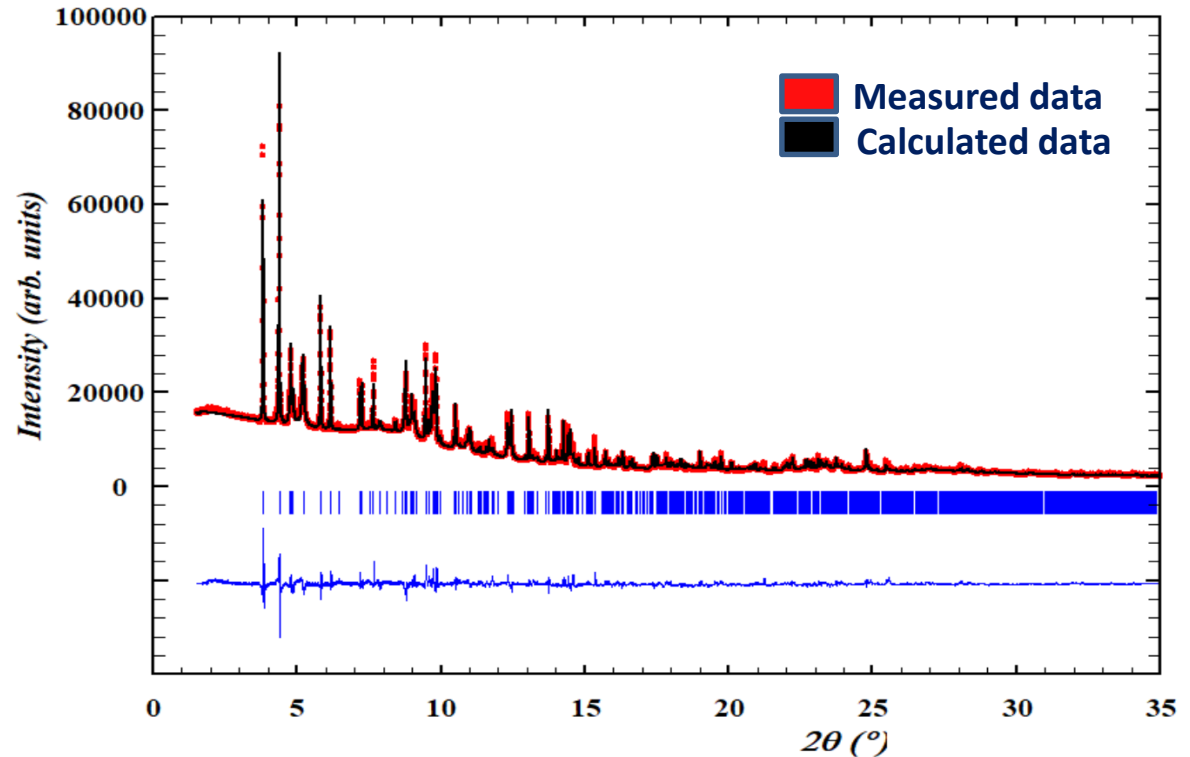
# Limitations of X-Ray diffraction

## Single Crystal X-Ray Diffraction:

Difficulties in growing crystals to sufficient size

## Powder X-Ray Diffraction:

Reflection overlapping due to mixtures and structure complexity



# Using electrons

Electrons interact strongly with matter,  
 an extraordinary advantage for studying nano-sized materials.



Neutrons  
 1 mm



Laboratory X-Rays  
 10-100  $\mu\text{m}$



Synchrotron X-Rays  
 1-100  $\mu\text{m}$



Transmission Electron  
 Microscope (TEM)  
 100-500 nm

**Minimal crystal size necessary for structure solution**

Ideal for  
 Complex structures  
 Nanocrystals,  
 Polyphased powder  
 ...

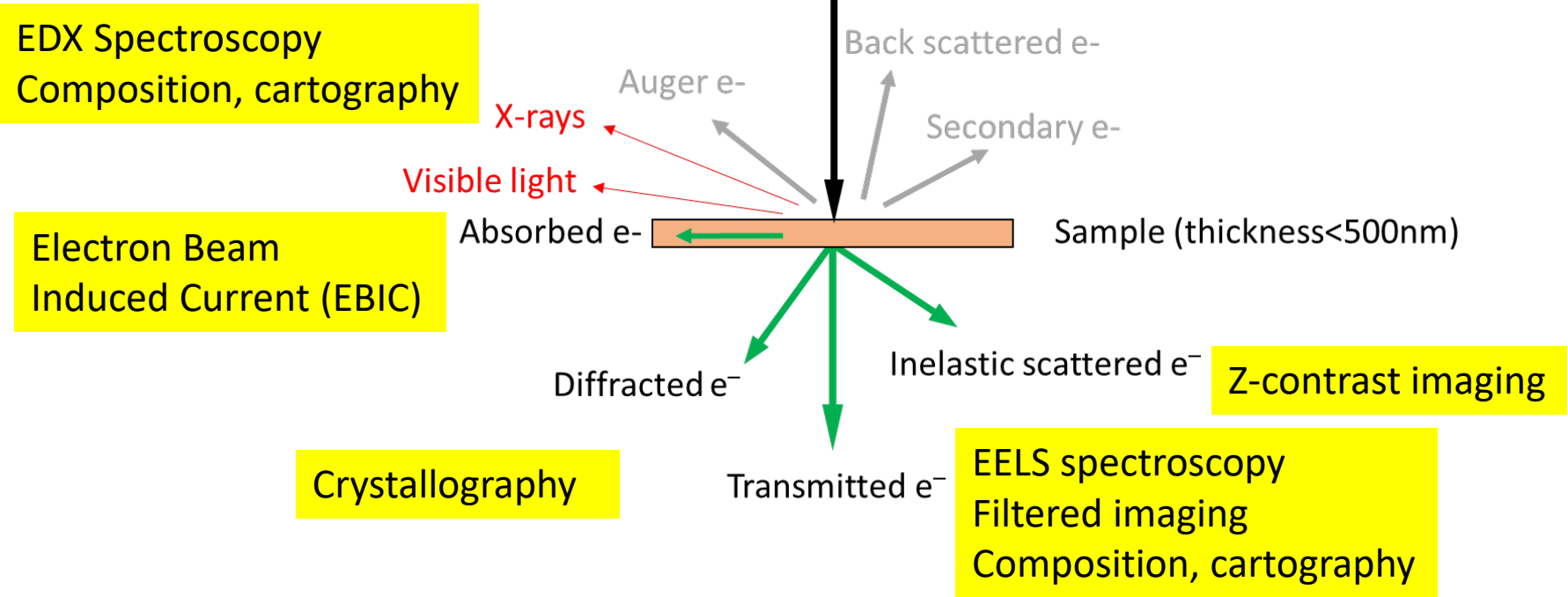
## **OUTLINE**

- ★ Introduction
- ★ **Particularities of transmission electron microscopy**
- ★ **Particularities of electron diffraction**
- ★ **3D electron diffraction: data collection and treatment**
- ★ **3D electron diffraction at Institut Néel**
- ★ **Conclusion**

# Particularities of transmission electron microscopy (TEM)

# Particularities of TEM: inelastic interactions

## Strong interaction of electrons with matter (with nucleus and electronic 'clouds')

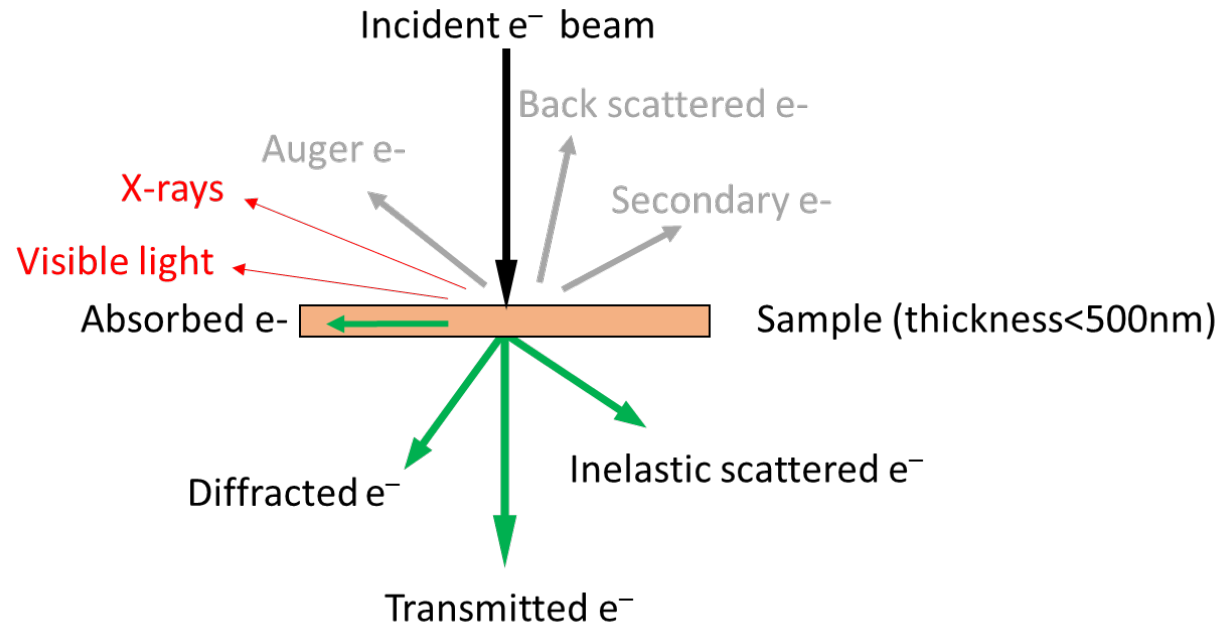


Many signals can be used

TEM is a **powerful tool** for sample characterization

## Particularities of TEM: inelastic interactions

Strong interaction of electrons with matter  
(with nucleus and electronic 'clouds')

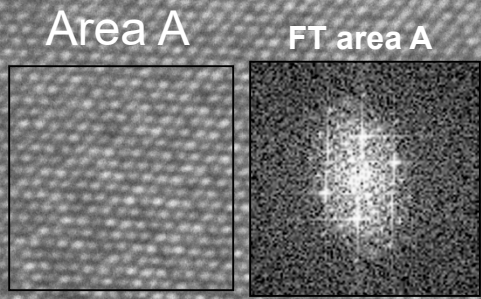


Inelastic interactions affect the structure and/or the chemistry  
 $\Rightarrow$  specimen is not representative of its parent material

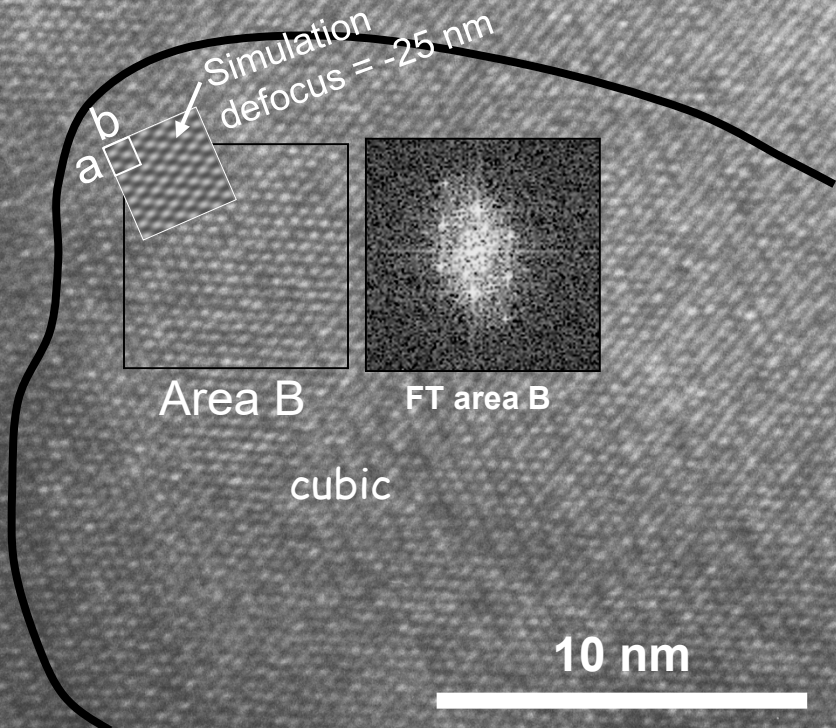
**Observing the specimen can change it**



TEM, 200 kV

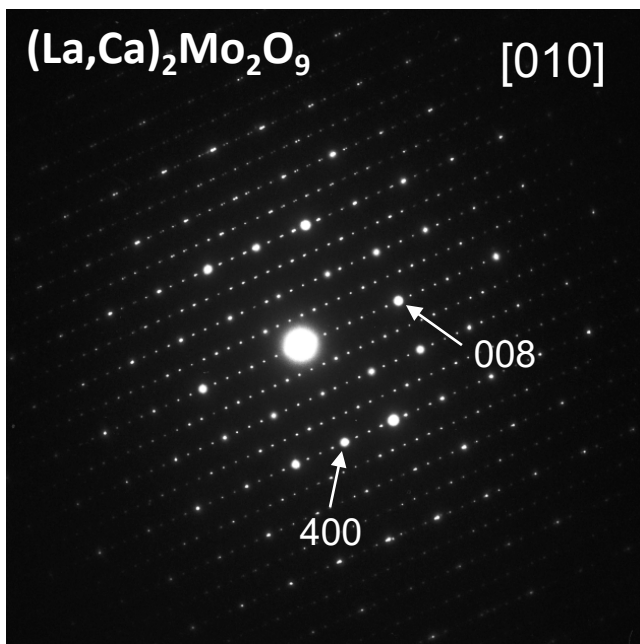


monoclinic



# Inelastic interactions: phase transition

Experimental conditions : selected area electron diffraction, room temperature, 200 kV



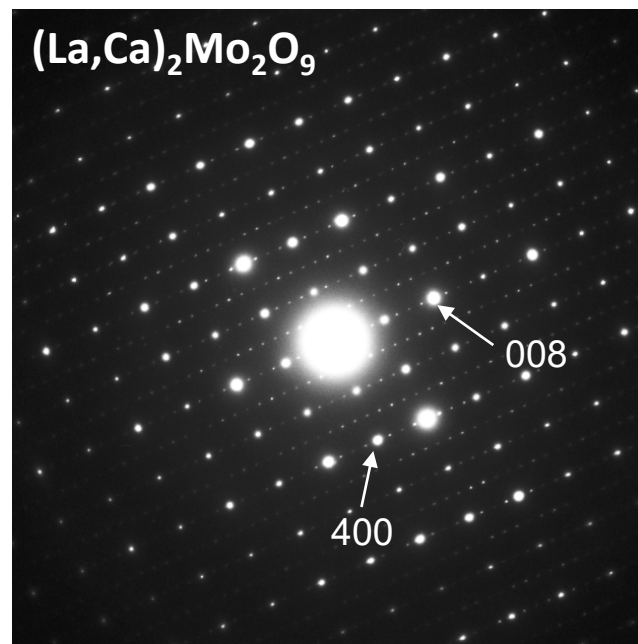
Superstructure x2 x3 x4

$\alpha$ -La<sub>2</sub>Mo<sub>2</sub>O<sub>9</sub> (Monoclinique P2<sub>1</sub>)  
 $a = 14.325(3) \text{ \AA}$ ,  $b = 21.482(4) \text{ \AA}$ ,  $c = 28.585(6) \text{ \AA}$   
 $\beta = 90.40(3)^\circ$

A few minutes  
Irradiation



Phase transition

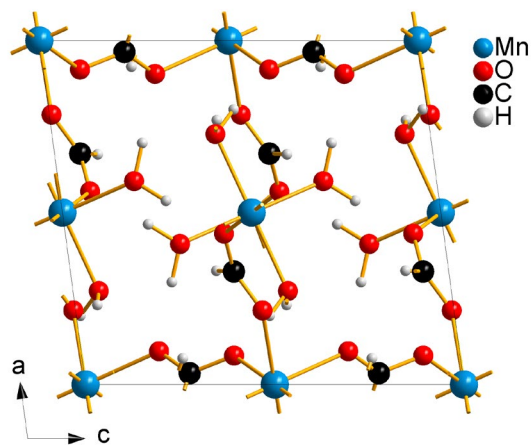


$\alpha$ -La<sub>2</sub>Mo<sub>2</sub>O<sub>9</sub> (Monoclinique P2<sub>1</sub>)  
 +  
 $\beta$ -La<sub>2</sub>Mo<sub>2</sub>O<sub>9</sub> (Cubique P2<sub>1</sub>3)  
 $a = 7.2351(1) \text{ \AA}$

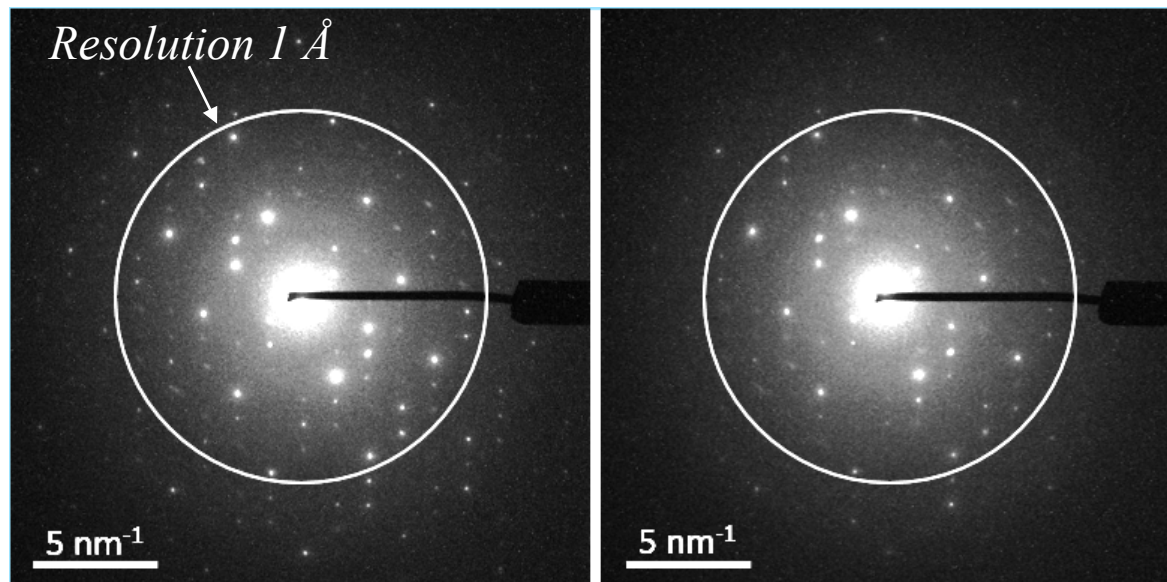


# Inelastic interactions: cristallinity loss

Manganese formate  
 $[Mn(HCOO)_2(H_2O)_2]_{\infty}$



Diffraction fading



*At the beginning  
of the data acquisition*

*After the data acquisition  
(same orientation)*

**Maximal irradiation dose:  $0.15 \text{ e}^-/\text{\AA}^2$**   
 300 kV, Crystal size:  $0.7 \mu\text{m} * 0.9 \mu\text{m}$

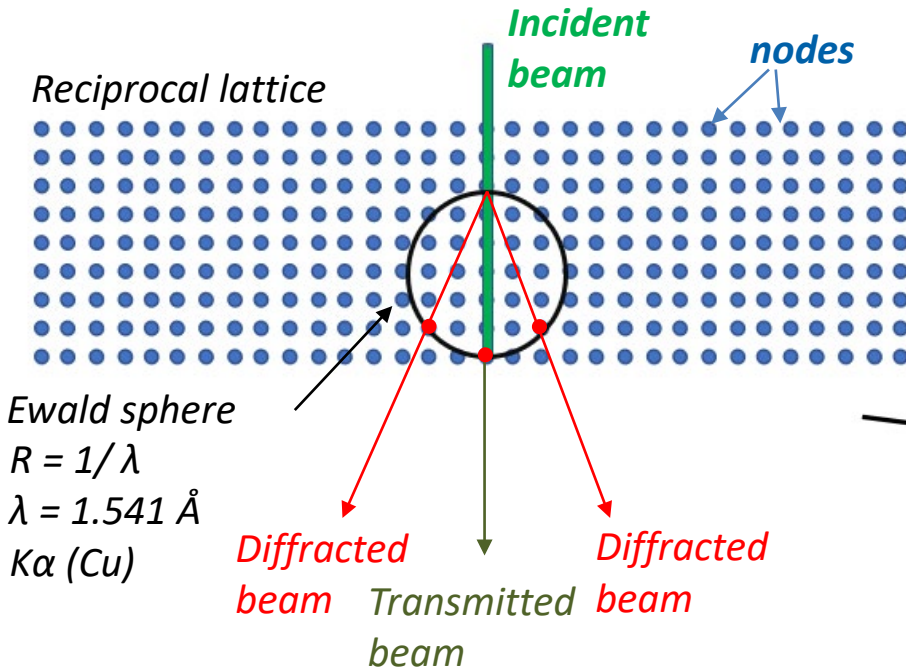
# Particularities of Electron diffraction

# Particularities of electron diffraction: rods and Ewald sphere

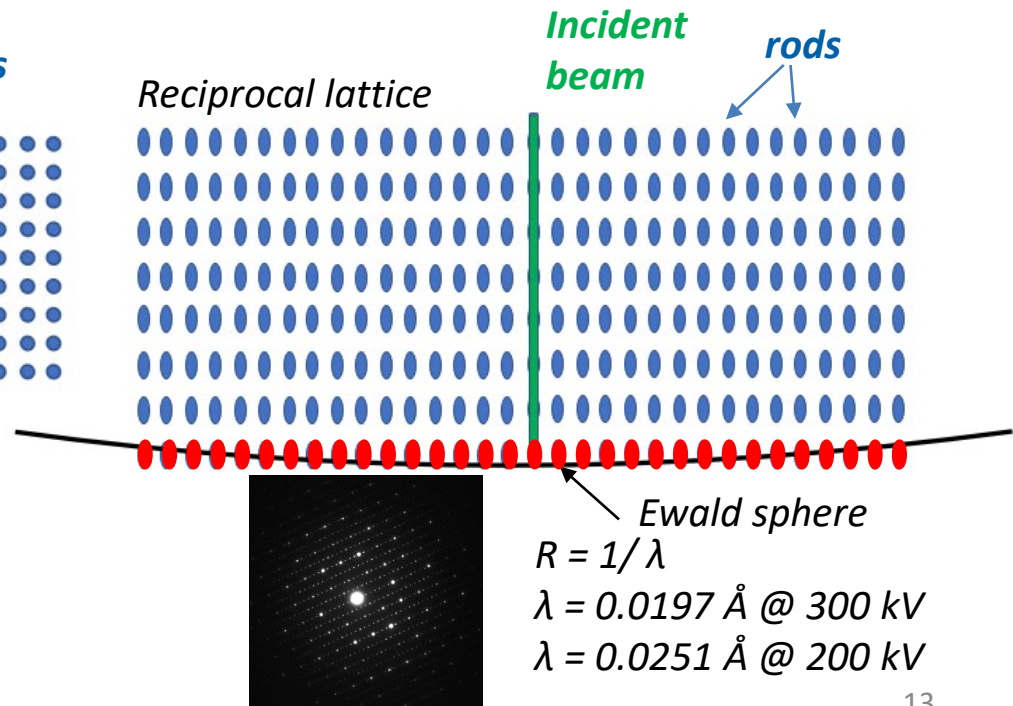
- **Thin** specimen => spatial extension of nodes (**rods**)
- **Small** wavelength => Ewald sphere **intersects** a large number of rods

$$\text{Bragg's law: } 2 d_{hkl} \sin \theta = n \lambda$$

**X-ray diffraction**

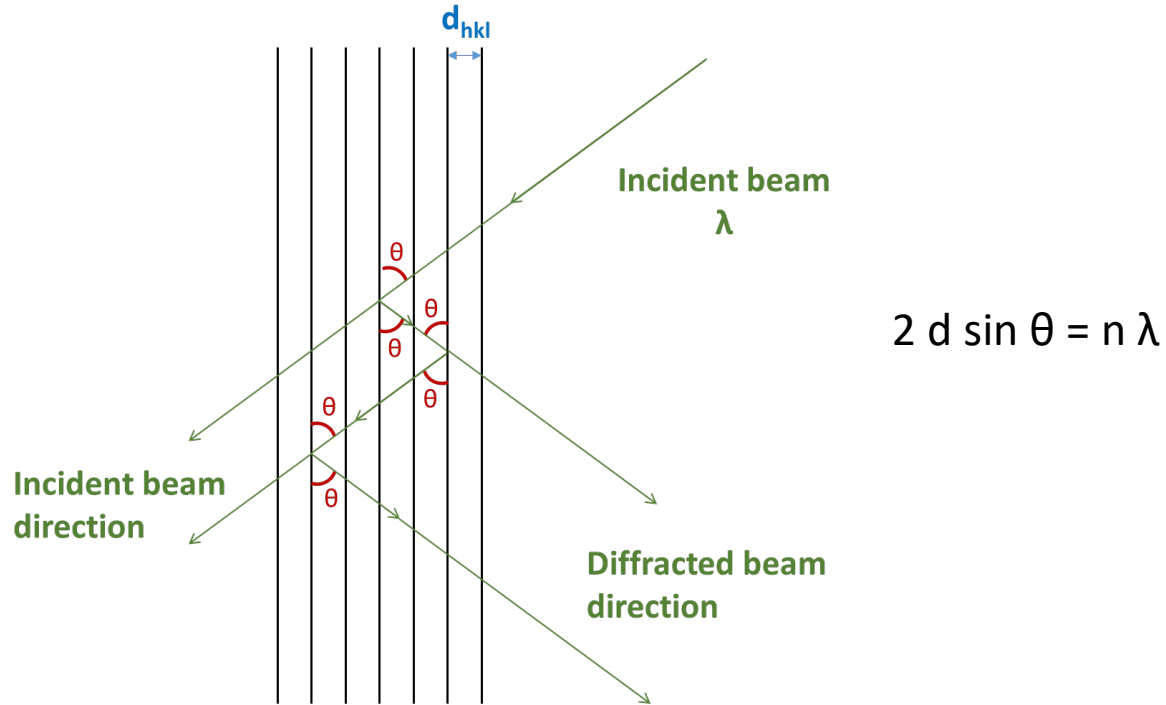


**Electron diffraction**

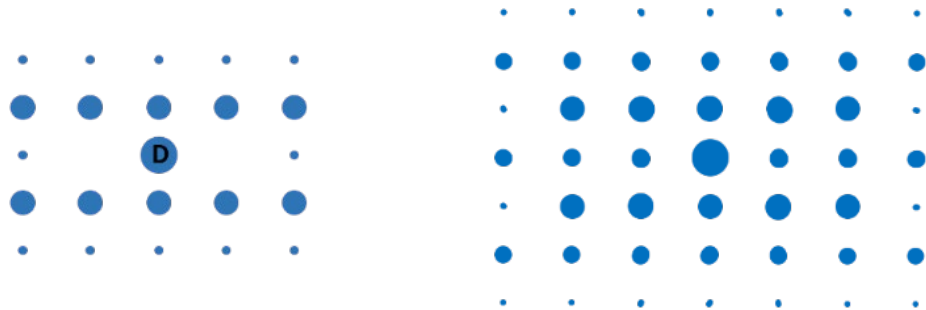


Figures: adapted from Emre YORUK, Institut Néel Grenoble and FZU Prague (PhD manuscript)

# Particularities of electron diffraction: dynamical effects



$I(hkl) \propto |F(hkl)|^2$



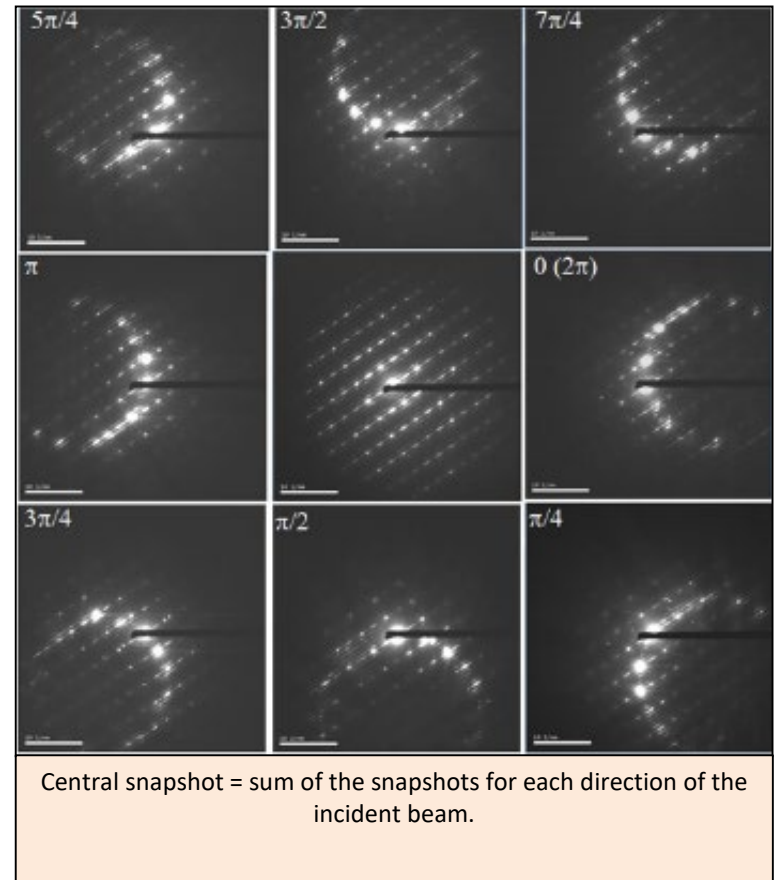
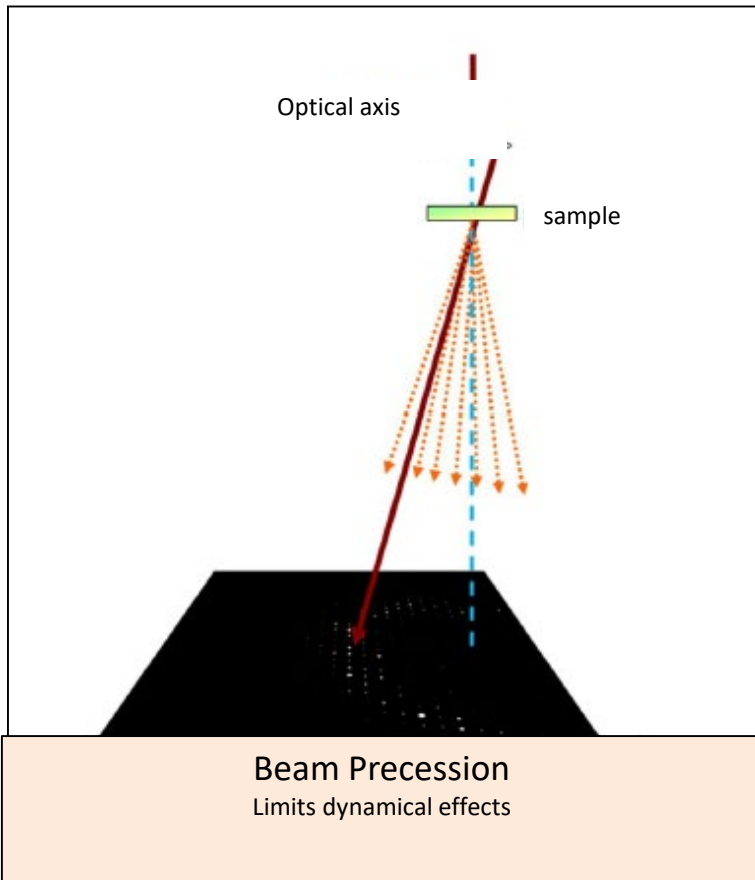
Kinematical diffraction pattern      Dynamical diffraction pattern

Relative *intensities are modified* due to dynamical diffraction

# Electron Diffraction: Recent Breakthroughs

## ❑ Precession Electron Diffraction (1994)

Vincent & Midgley, *Ultramicroscopy* 53 (3), 1994



# Electron Diffraction: Recent Breakthroughs

❑ **Precession Electron Diffraction (1994)**

*Vincent & Midgley, Ultramicroscopy 53 (3), 1994*



**Dynamical effects reduced**

❑ **Automated Diffraction Tomography (2007)**

*Kolb et al., Ultramicroscopy 107 (6-7), 2007*

❑ **Precession Electron Diffraction Tomography (2009)**

*Mugnaioli et al., Ultramicroscopy 109 (6), 2009*

❑ **Continuous Rotation Electron Diffraction (2013)**

*Nederlof et al., Acta Cryst. D 69 (7), 2013. Etc...*



**Faster data acquisition:  
3D Electron Diffraction**

❑ **Dynamical refinement (2015)**

*Palatinus et al., Acta Cryst. A 71 (2), 2015*



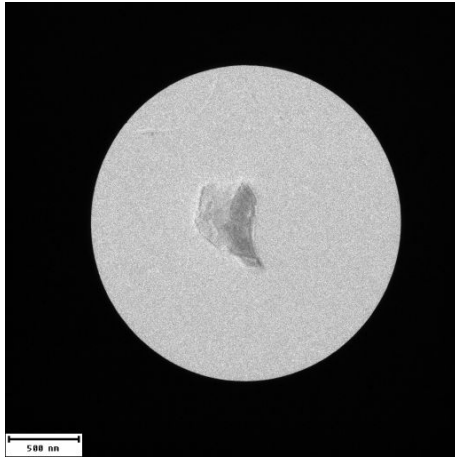
**Consideration of dynamical diffraction in data processing**



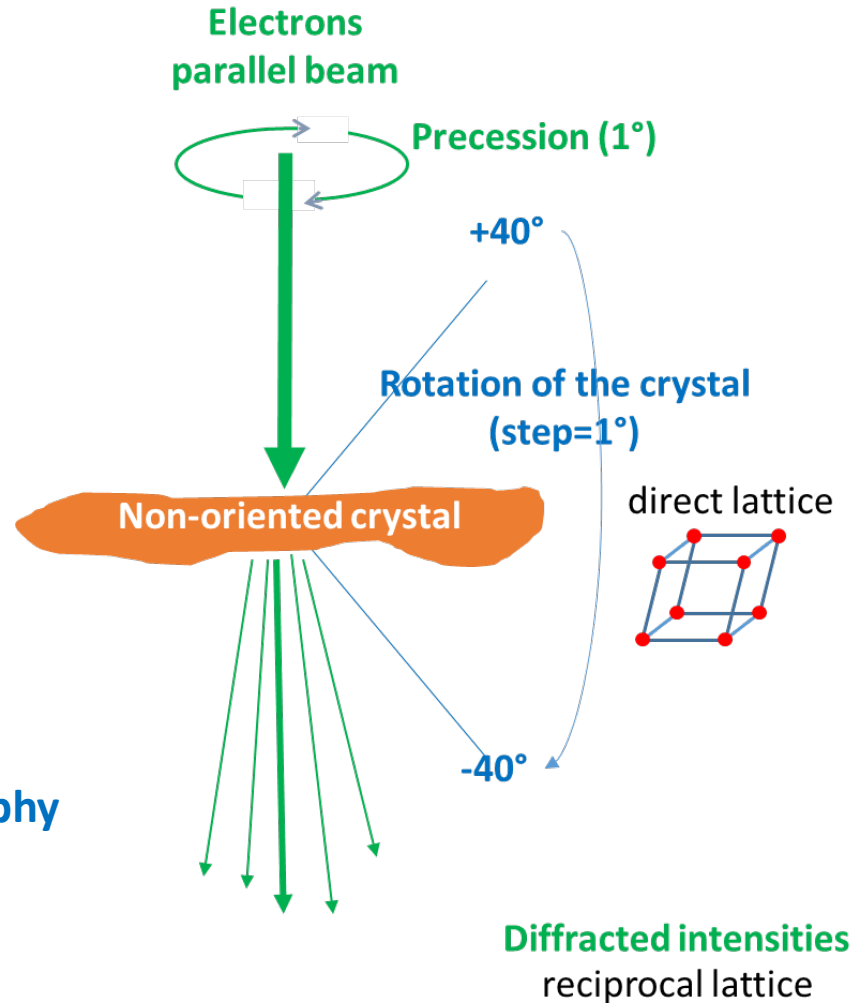
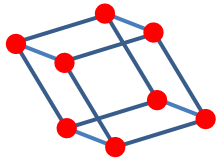
# **3D electron diffraction: Data collection and treatment**

# Data collection: Low Dose Electron Diffraction Tomography

S. Kodjikian and H. Klein, *Ultramicroscopy*, 200, 2019, pp. 12-19



Non-oriented crystal



## Low Dose –Electron Diffraction Tomography

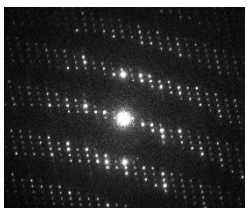
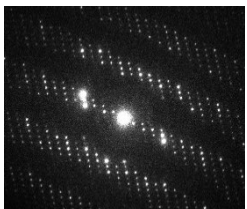
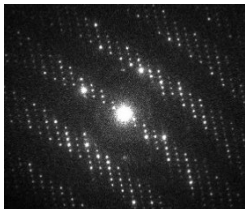
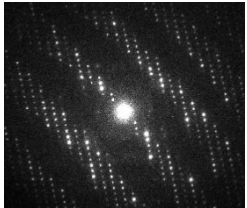
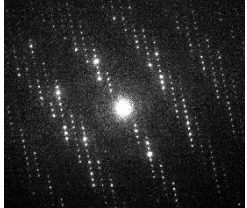
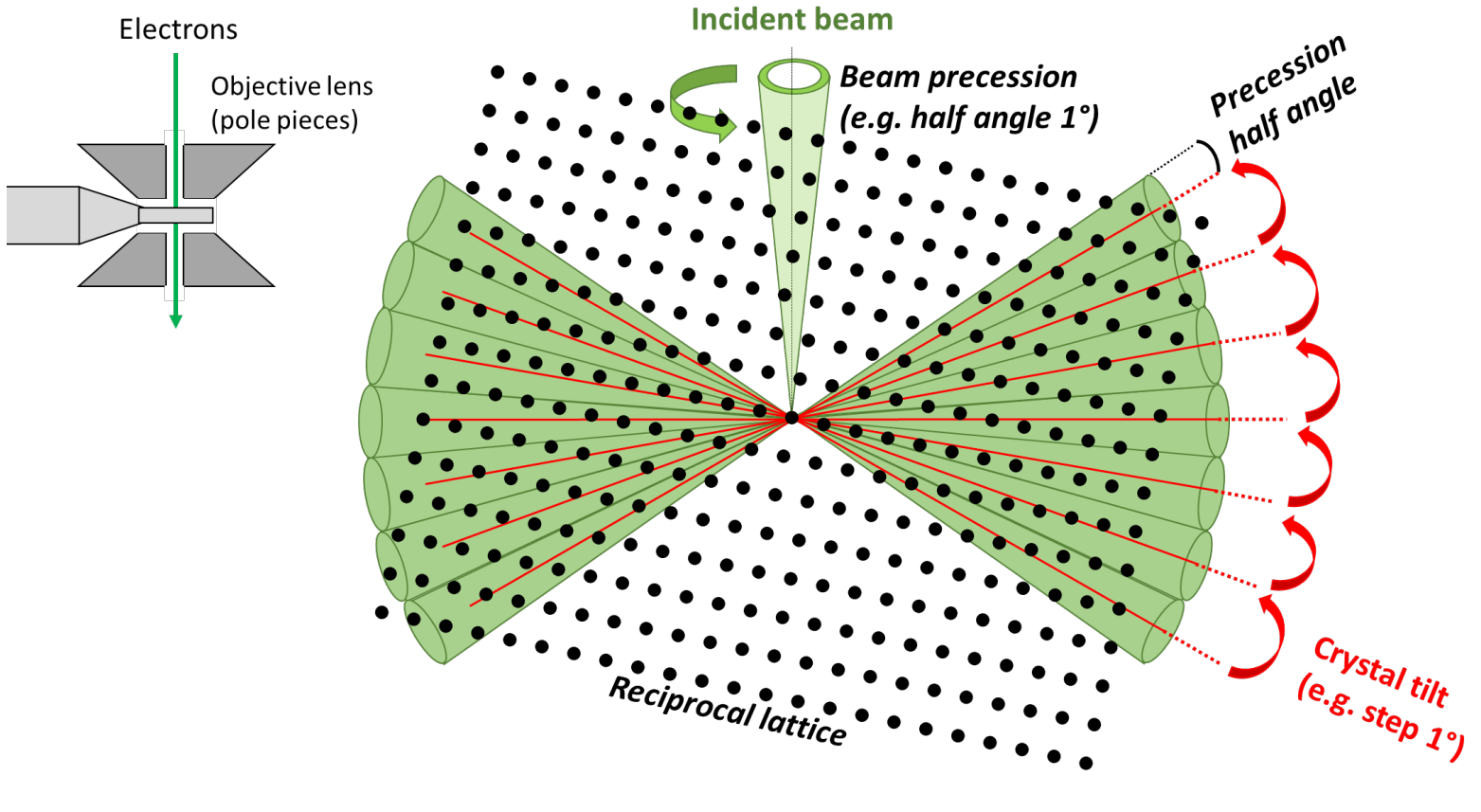
- Step by step (step = 1°)
- Beam precession ( half-angle = 1°)
- Beam blanking between two frames.

# Data collection: example of LD-EDT method

S. Kodjikian and H. Klein, *Ultramicroscopy*, 200, 2019, pp. 12-19

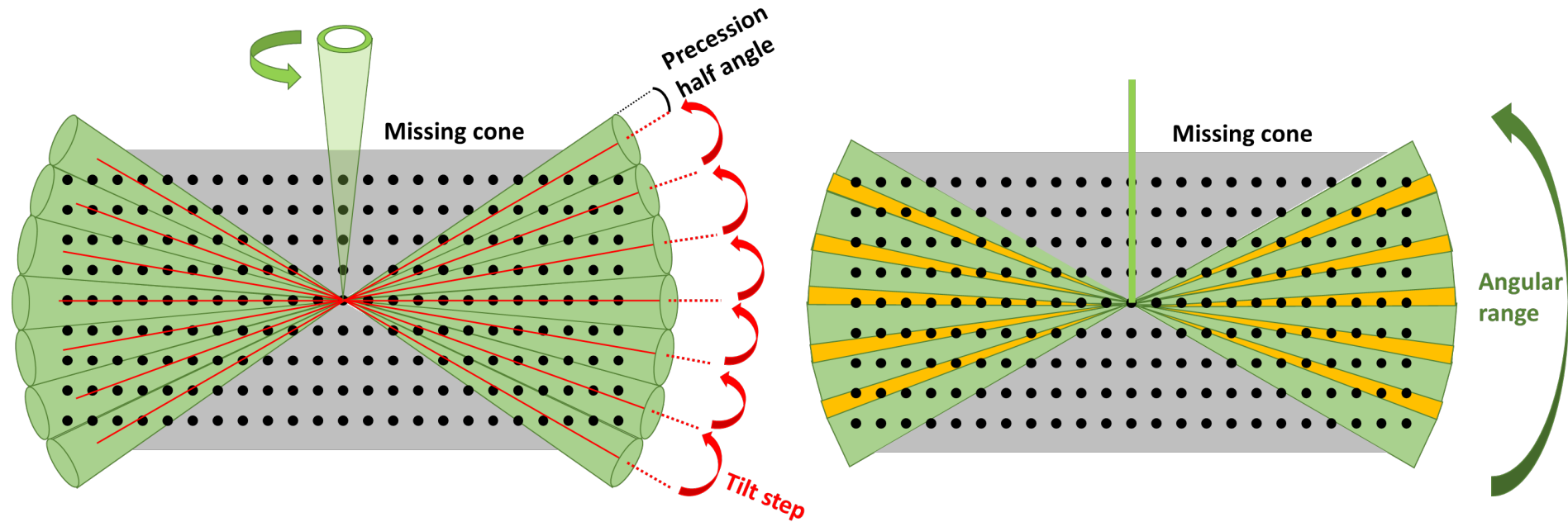


... 3D Reconstruction of the reciprocal lattice



## 3D Electron Diffraction – data collection

3D ED: Collecting ED data from a single crystal while tilting it around an arbitrary axis



### Step-wise rotation, with beam precession (ADT, PEDT, LD-EDT)

- ✓ Possibility to re-center the sample
- ✓ Time consuming (10 min to 45 min)
- ✓ Requires beam precession

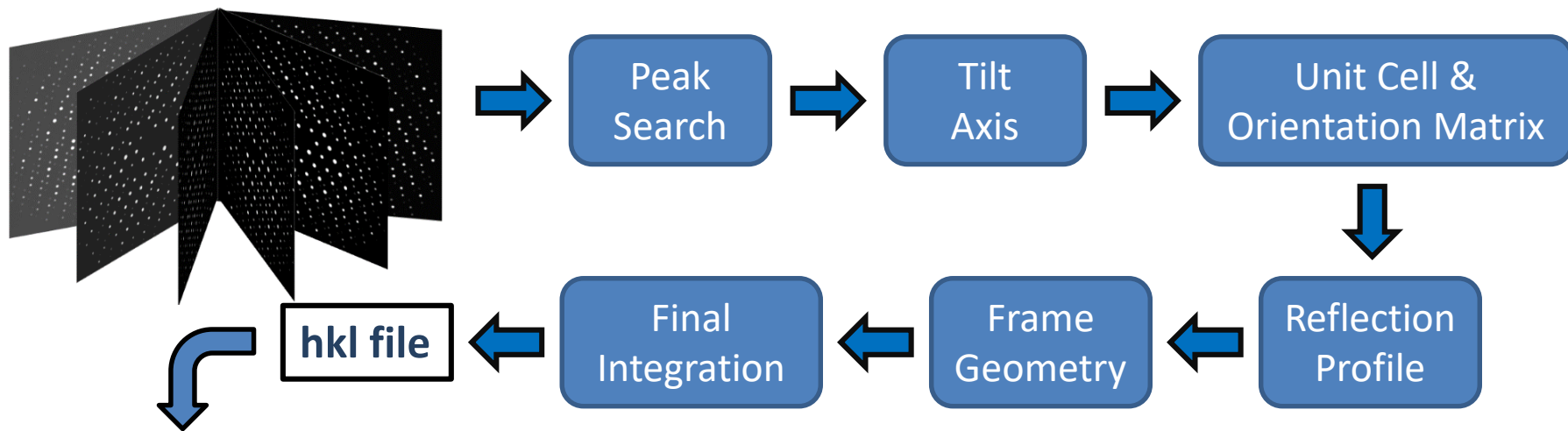
=> **LOW DOSE** (beam blanking)

### Continuous rotation (cRED, MicroED)

- ✓ Very fast : 30 s to 5 min
- ✓ Requires
  - very stable goniometer & sample holder
  - fast detector

=> **LOW DOSE**

# 3D Electron Diffraction – data treatment



Phasing by

- Direct Methods
- Charge Flipping
- Molecular Replacement

Kinematical + Dynamical Refinement

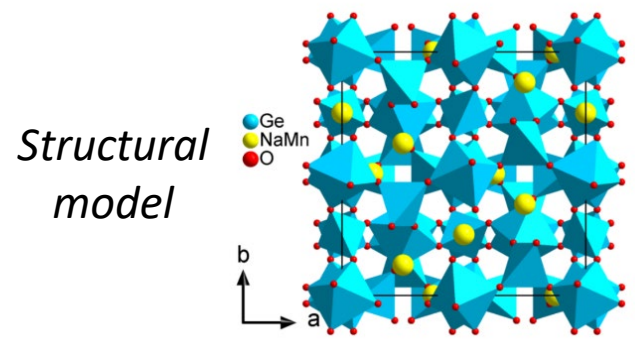
**Final Structure**

### Kinematical refinement

- $I_{hkl} \propto |F_{hkl}|^2$
- One intensity per reflection

### Dynamical refinement

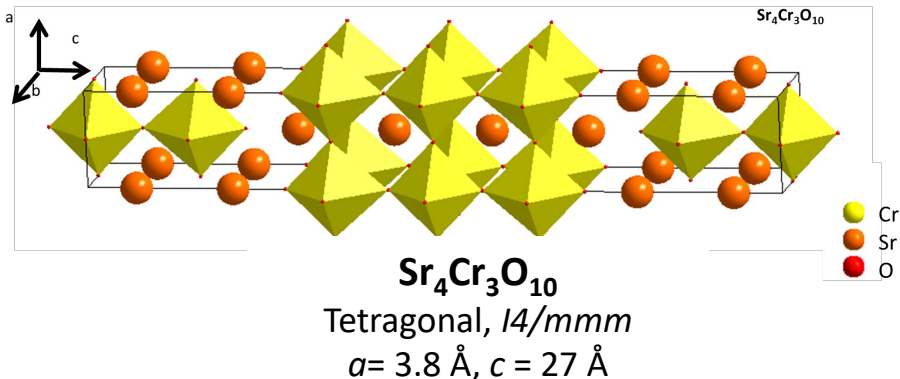
- $I_{hkl} \propto \text{multiple } |F|^2$
- Reflections considered frame by frame



# 3D ED @ Institut Néel

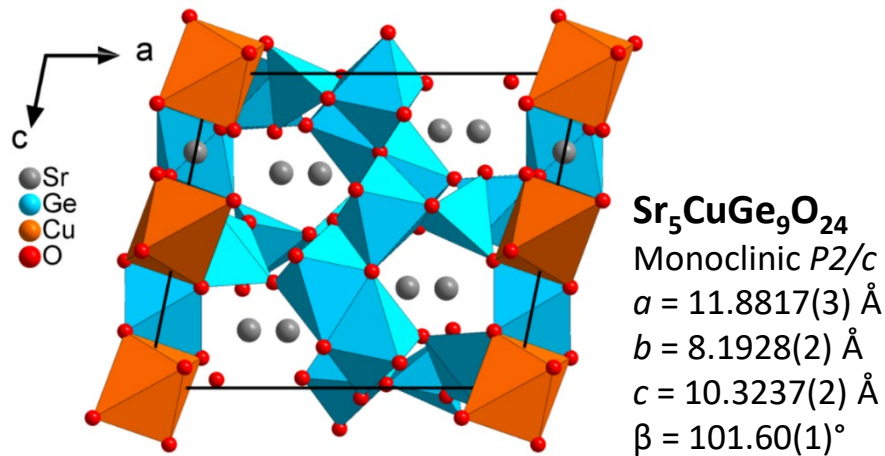
# Some unknown structures solved at Institut Néel

## Superconductivity



*J. Jeanneau et al. Journal of Solid State Chemistry 251 (2017) 164-169*

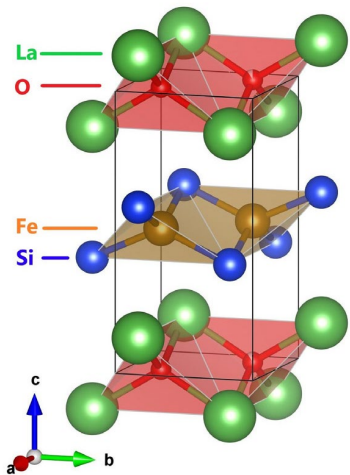
## Magnetic properties / multiferroism



*H. Klein et al. Acta Cryst. (2020) B76, (5), pp.727-732*

Synthesized at HP/HT  
 3 Ge environments:  
 Tetrahedra, octahedra,  
 five-fold pyramid

## Superconductivity, T<sub>c</sub> = 10 K



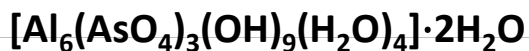
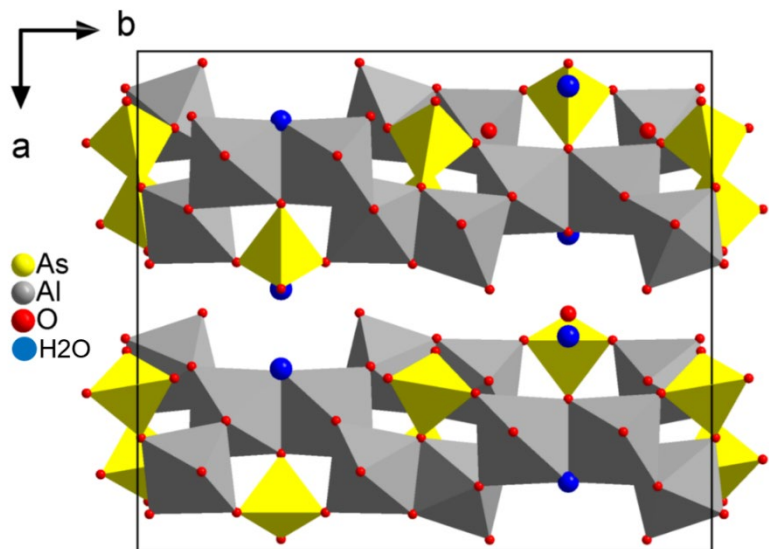
**LaFeSiO<sub>1-δ</sub>**  
*P4/nmm*  
 $a = 4.1085(4) \text{ \AA}$ ,  
 $c = 8.132(2) \text{ \AA}$

*M.F. Hansen et al. Npj Quantum Materials, 2022, 7 (1), pp.86*



# Some unknown sensitive materials solved at Institut Néel

## Mineral

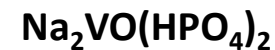
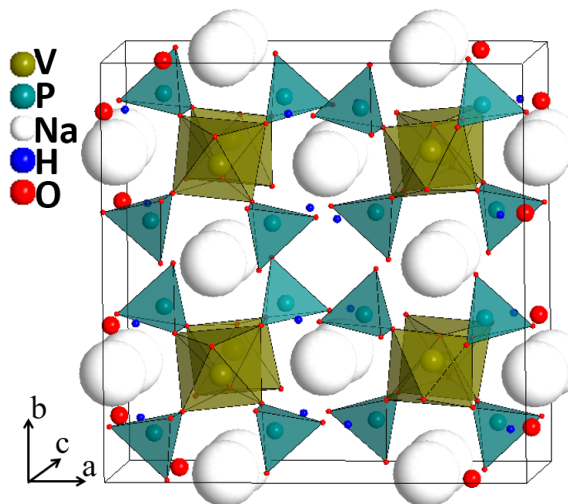


*Pnma*,  $a = 15.4 \text{ \AA}$ ,  $b = 17.7 \text{ \AA}$ ,  $c = 7.81 \text{ \AA}$

Total dose :  $3 \text{ e}^-/\text{\AA}^2$

*I.E. Grey et al. Mineralogical Magazine (2020), 84, 608–615*

## Na-ions batteries



*Iba2*

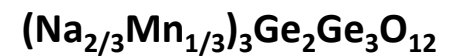
$a = 13.86852(19) \text{ \AA}$

$b = 13.7985(2) \text{ \AA}$

$c = 7.47677(9) \text{ \AA}$

*C. Lepoittevin et al, Dalton Trans., 2021, 50, 9725-9734*

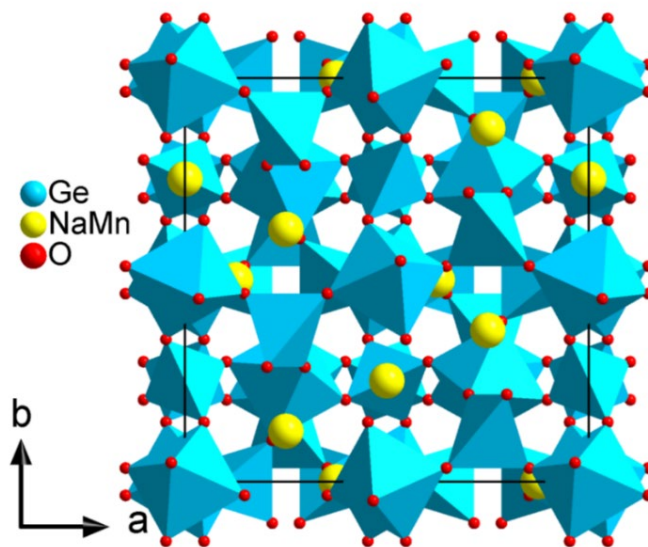
## Quasi-1D magnetism, Multiferroism



*Cubic Ia-3d*,  $a = 11.9860(3) \text{ \AA}$

Total dose :  $0.13 \text{ e}^-/\text{\AA}^2$

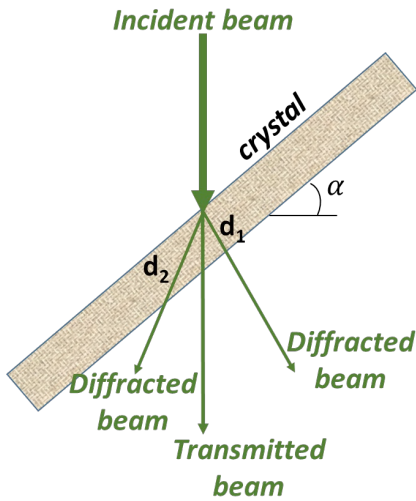
*H. Klein et al. Symmetry 2022, 14 (2), 245*





# 3D Electron Diffraction at Institut Néel: improvements Dose Symmetric Electron Diffraction Tomography (DS-EDT)

★ When the crystal is tilted, diffracted beams experience different apparent thicknesses.

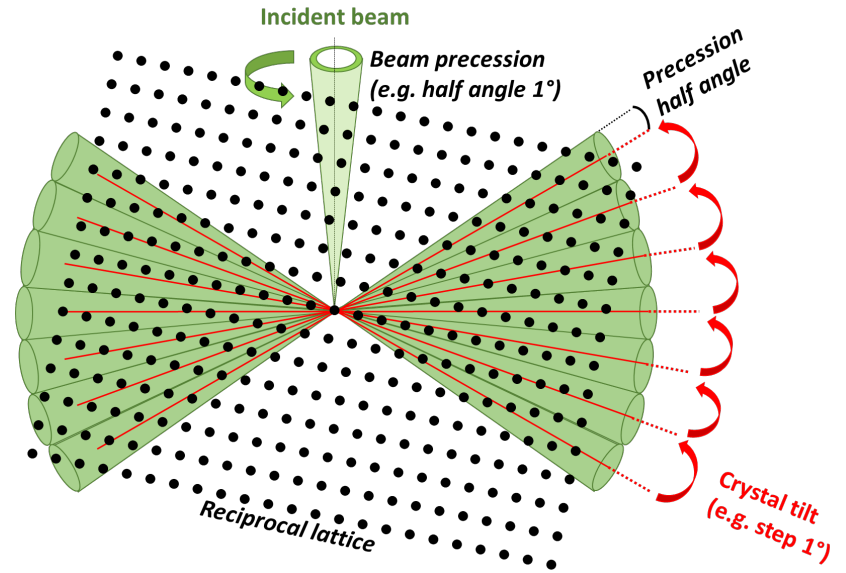


Differences in absorption and dynamical effects

### Low tilt

- ⇒ low apparent thickness
- ⇒ High resolution

★ Typical step by step tomography in 3D ED



The initially recorded high tilt frames are the least damaged, while the damage accumulates throughout the acquisition.

- ⇒ Record low tilt frames first
- Minimum beam damage and high resolution

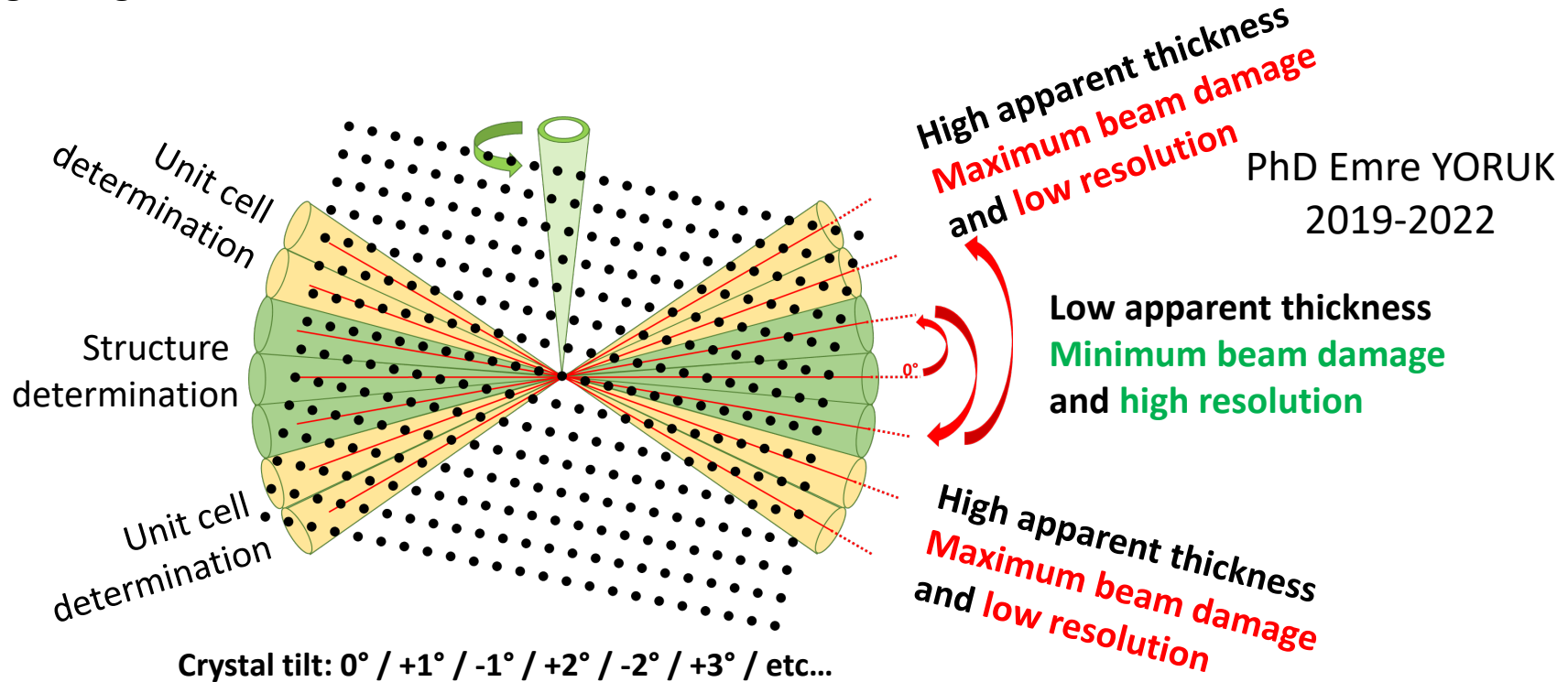
PhD Emre YORUK  
2019-2022

# 3D Electron Diffraction at Institut Néel: improvements

## Dose Symmetric Electron Diffraction Tomography (DS-EDT)

E. Yoruk et al. Ultramicroscopy 16 sept. 2023, DOI: [10.1016/j.ultramic.2023.113857](https://doi.org/10.1016/j.ultramic.2023.113857)

- ★ Data collection begins at low tilt, and alternates between increasingly positive & negative tilts
- ★ **Low tilt frames with high-resolution information are least irradiated**
- ★ Damaged high tilt frames used to determine the unit cell and the orientation matrix



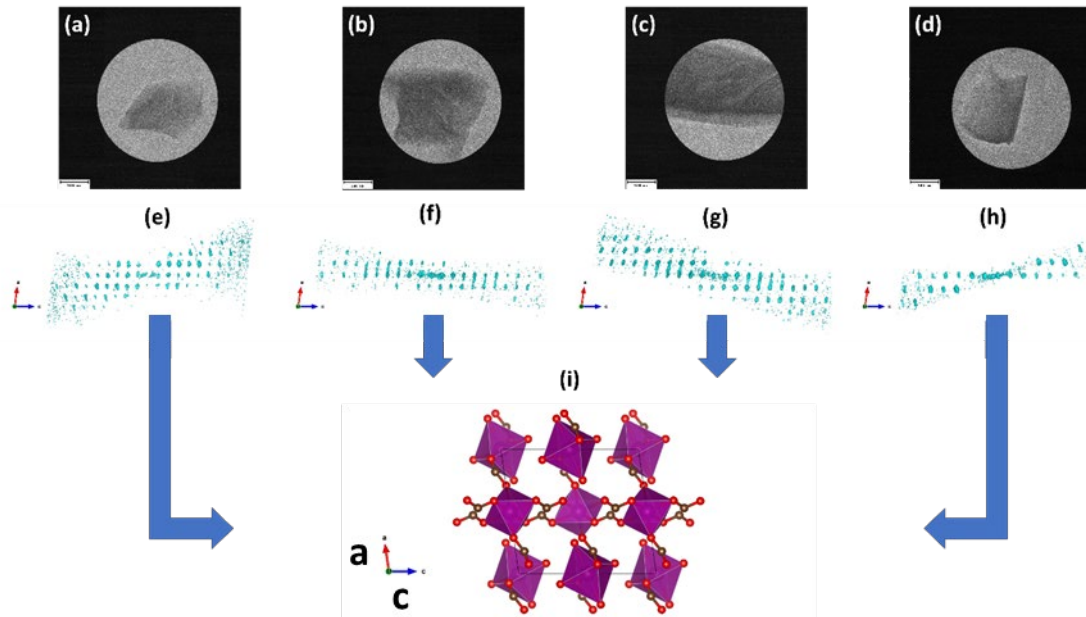
- ★ Only low tilt frames are used for structure determination, preserving data from beam damage  
→ **Lower dose per particle without lowering the signal to noise ratio**

# 3D Electron Diffraction at Institut Néel: improvements Dose Symmetric Electron Diffraction Tomography (DS-EDT)

E. Yoruk et al. *Ultramicroscopy* 16 sept. 2023, DOI: [10.1016/j.ultramic.2023.113857](https://doi.org/10.1016/j.ultramic.2023.113857)

★ Low tilt data from multiple crystals is merged for data completeness

Software  
Pets 2  
Jana 2020



**Example:**  
Mn formate  
 $\text{Mn}_2(\text{COOH})_2(\text{H}_2\text{O})_2$   
Monoclinic  $P2_1/c$   
 $a = 8.8263 \text{ \AA}$ ,  
 $b = 7.2247 \text{ \AA}$ ,  
 $c = 9.6305 \text{ \AA}$   
 $\beta = 97.689^\circ$

★ Structure solution and dynamical refinement

- **High accuracy** of the atom positions
- Possible using **only low-tilt data** from **multiple particles**, down to **ranges of +/- 5°** per particle

# 3D Electron Diffraction at Institut Néel: $\text{Ca}_2\text{MnO}_3\text{Cl}$

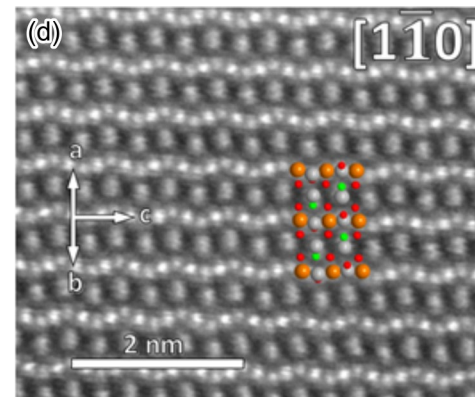
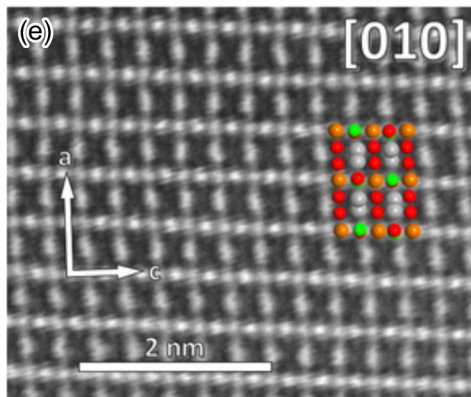
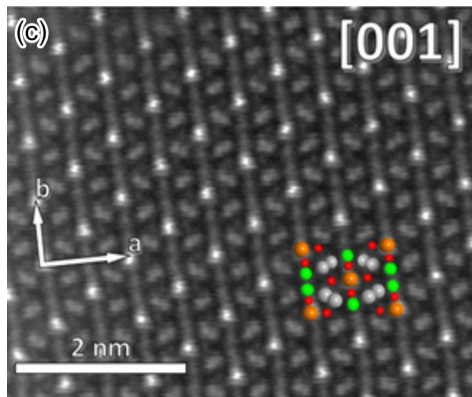
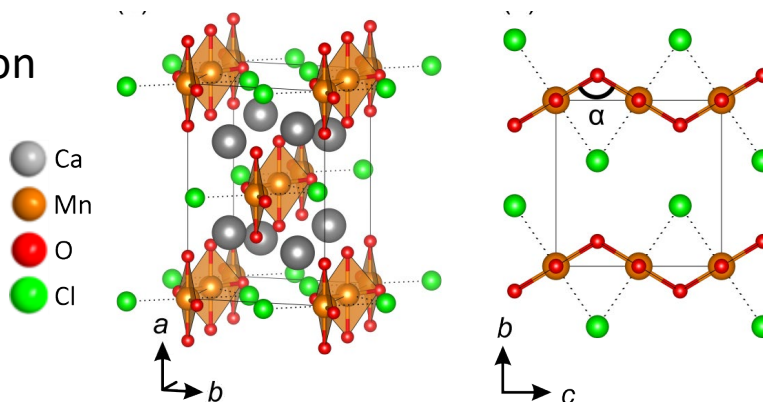
1-dimensional ferromagnetic chains

- ★ **TEM:** **Composition** from Energy Dispersive X-Ray Spectroscopy: Ca, Mn, O, Cl  
**Structural model** from 3D electron diffraction  
**Comparison** of the structural model with STEM HAADF images

★ **Powder X-Ray diffraction:**

Unit cell /space group confirmation  
 Structure refinement

*F. Denis Romero et al.,  
 Accepted by Journal of the  
 American Chemical Society*



STEM-HAADF Images: C. Lepoittevin

# Conclusion

## Conclusion

- ✓ 3D Electron diffraction suitable for a wide range of samples
  - Nanocrystals
  - Mixtures
  - Domains
  
- ✓ New developments in data collection and treatment
  - Detectors
  - Software
  
- ✓ New developments for beam sensitive materials
  - Preparation
  - Cryo
  - Low dose
  - New methods (merging data)

## Acknowledgments



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Rémy PHILIPPE

Pierre TOULEMONDE



**FZU**

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of the Czech  
Academy of Sciences



Ian GREY

CSIRO Melbourne



Valérie PRALONG

Crismat Caen



Dominique LUNEAU

LMI Lyon



**La Région**

Auvergne-Rhône-Alpes